

# Magnetron Sputtered Hydrogenated Amorphous Silicon Coatings for Einstein Telescope Mirrors

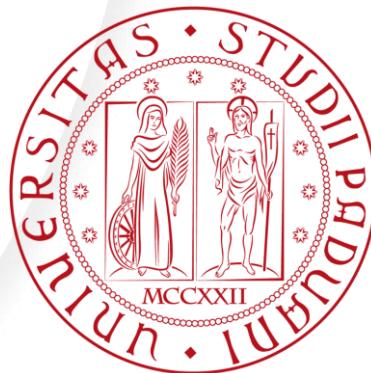
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# Introduction

## Amorphous silicon

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- Hydrogenation during deposition by radio frequency magnetron sputtering
- Compositional Analysis with annealing temperature

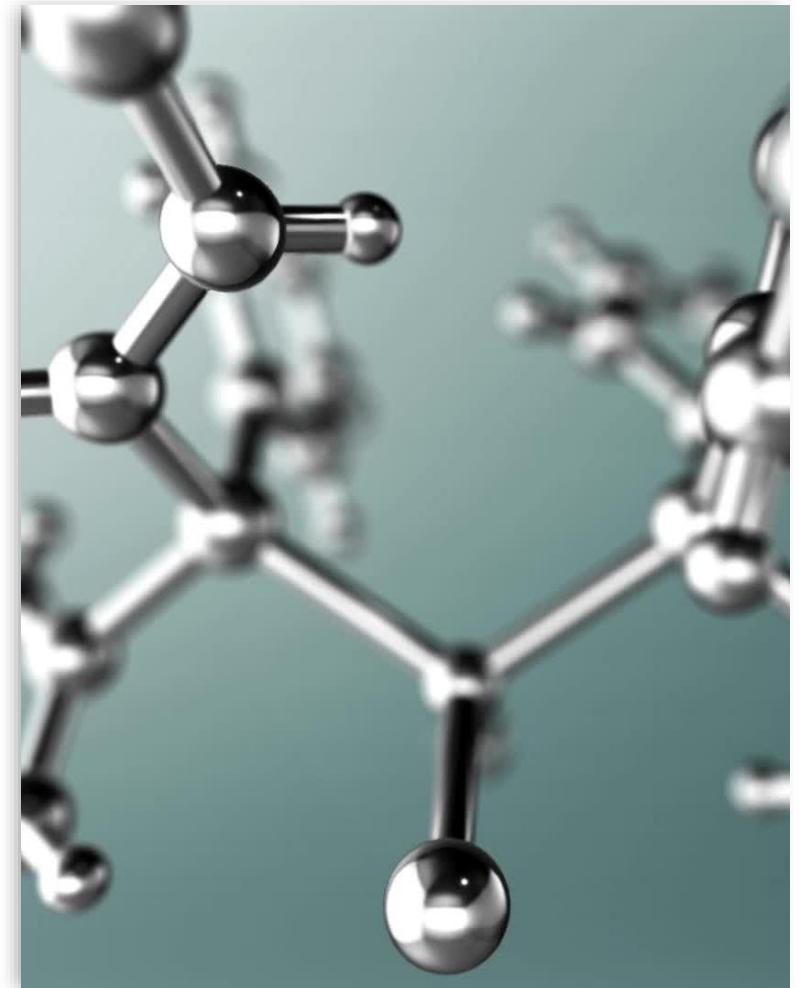
⑩ Maastricht University

- Refractive index vs annealing temperature
- Optical absorption vs annealing temperature

# Limitations of Amorphous Silicon

## High optical absorption

- Exceeds acceptable levels for gravitational-wave detectors
- Studies related the absorption in the NIR region of aSi to the dangling bonds (unpaired electron-spin density) [1]
- Hydrogenation can help reducing dangling bond and hence absorption [2]
- H terminates dangling bonds; rearrangement of hydrogen sub lattice can affect concentration and distribution in defects thus facilitates a-Si structural rearrangement



[1] Phys. Rev. Lett. 131, 256902

[2] Phys. Rev. Lett. 121, 191101

# Amorphous silicon coatings

## Deposition setup @LNL-INFN

Target Si 99.999%

Radio Frequency Magnetron Sputtering

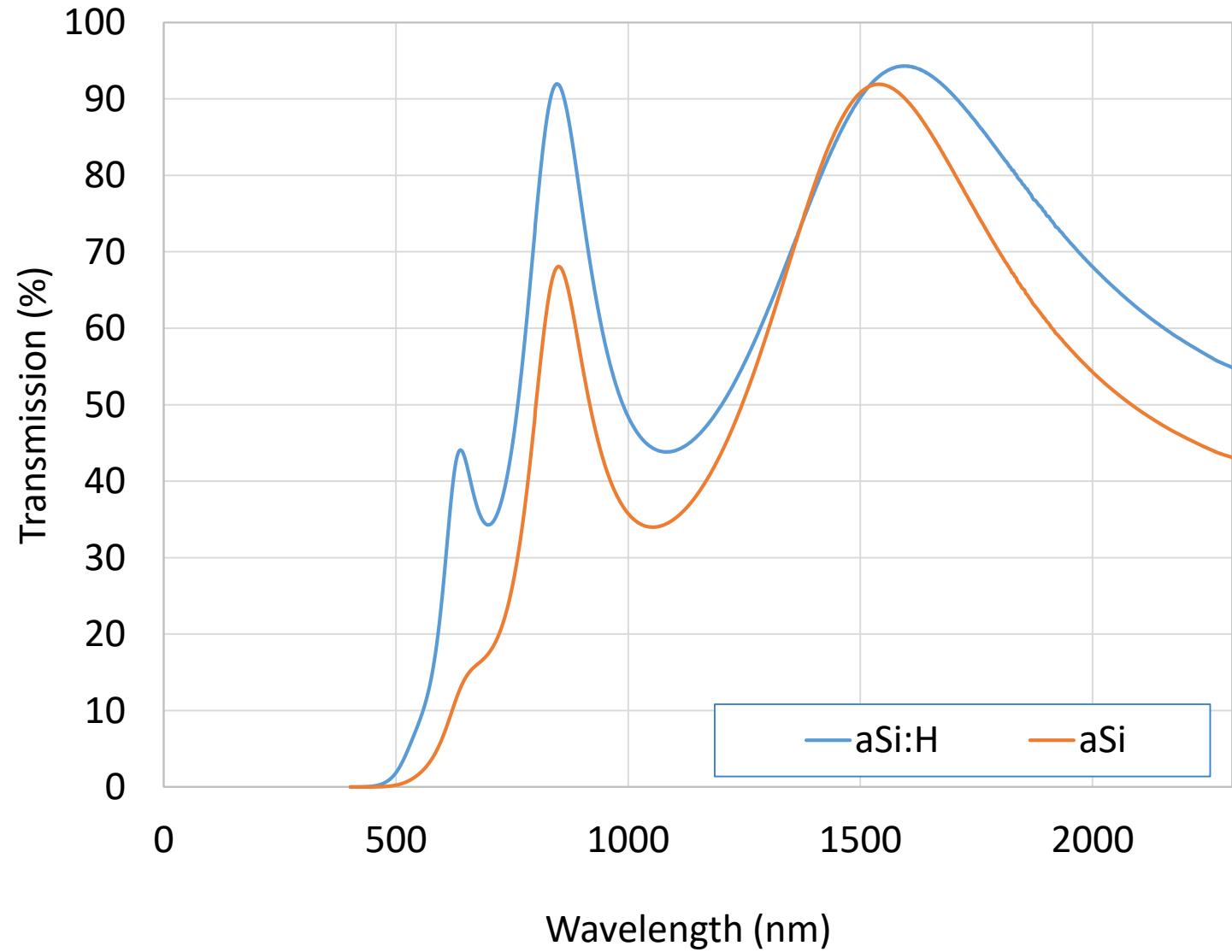
Deposition	aSi	aSi:H
Sputtering gas	Ar	Ar+5%H2
Substrates	2 SiO <sub>2</sub> disks (C7979), 2 Si wafer one side polished	2 SiO <sub>2</sub> disks (C7979), 2 Si wafer one side polished



# Optical Characterization

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# Transmission Spectra for as deposited coatings



# Transmission spectra

## Annealing steps

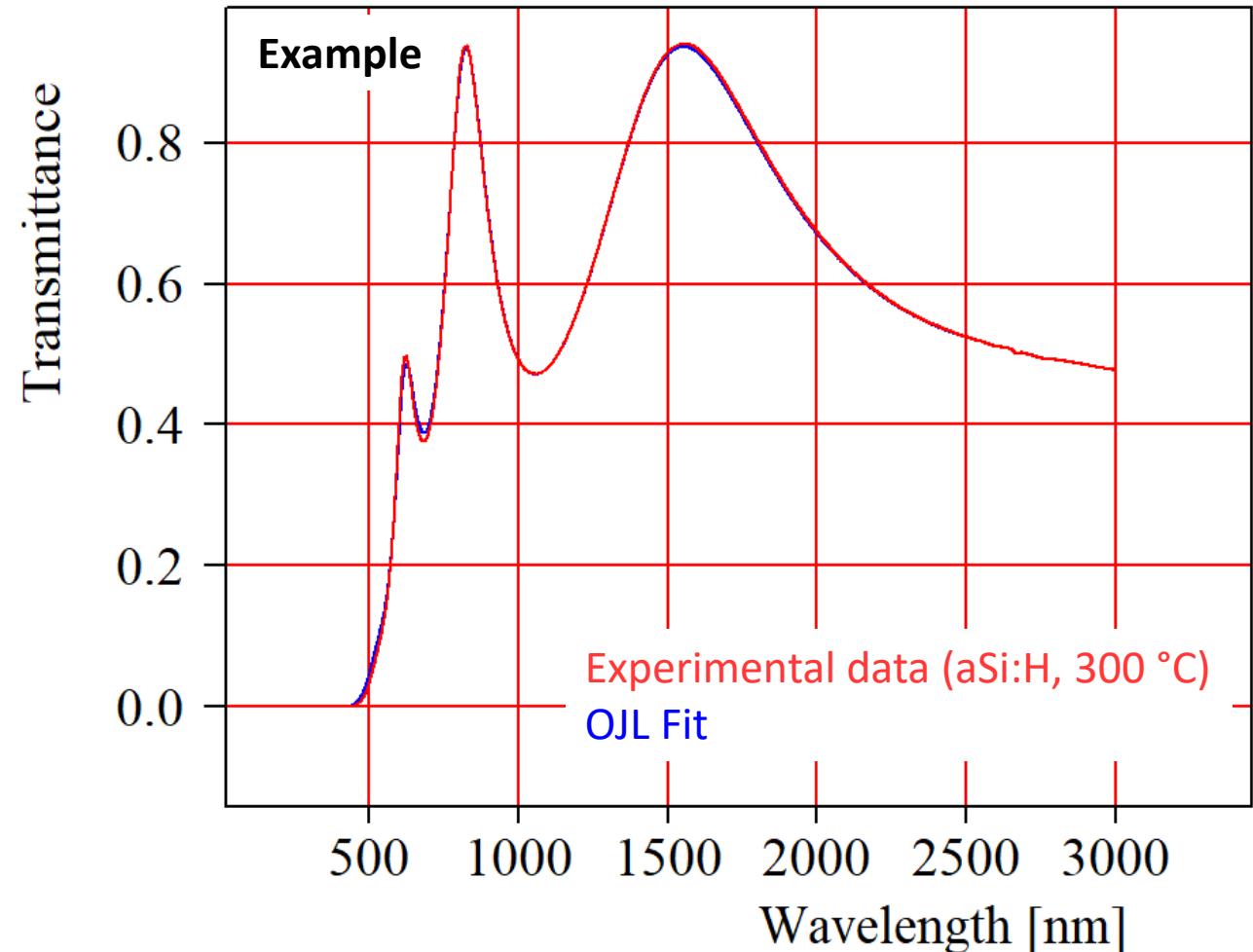
100°C steps from as deposited to 600 °C.

## Transmission spectra

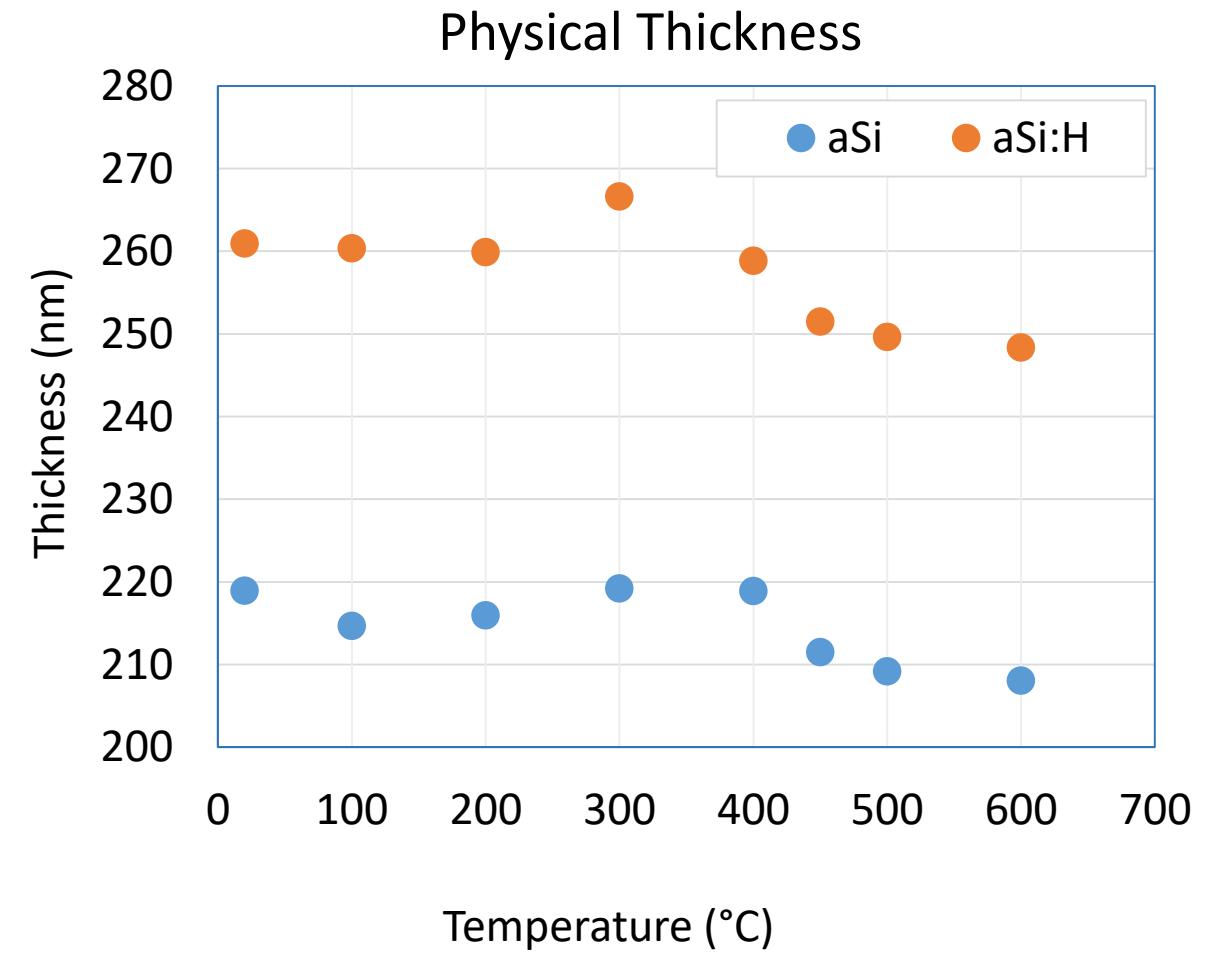
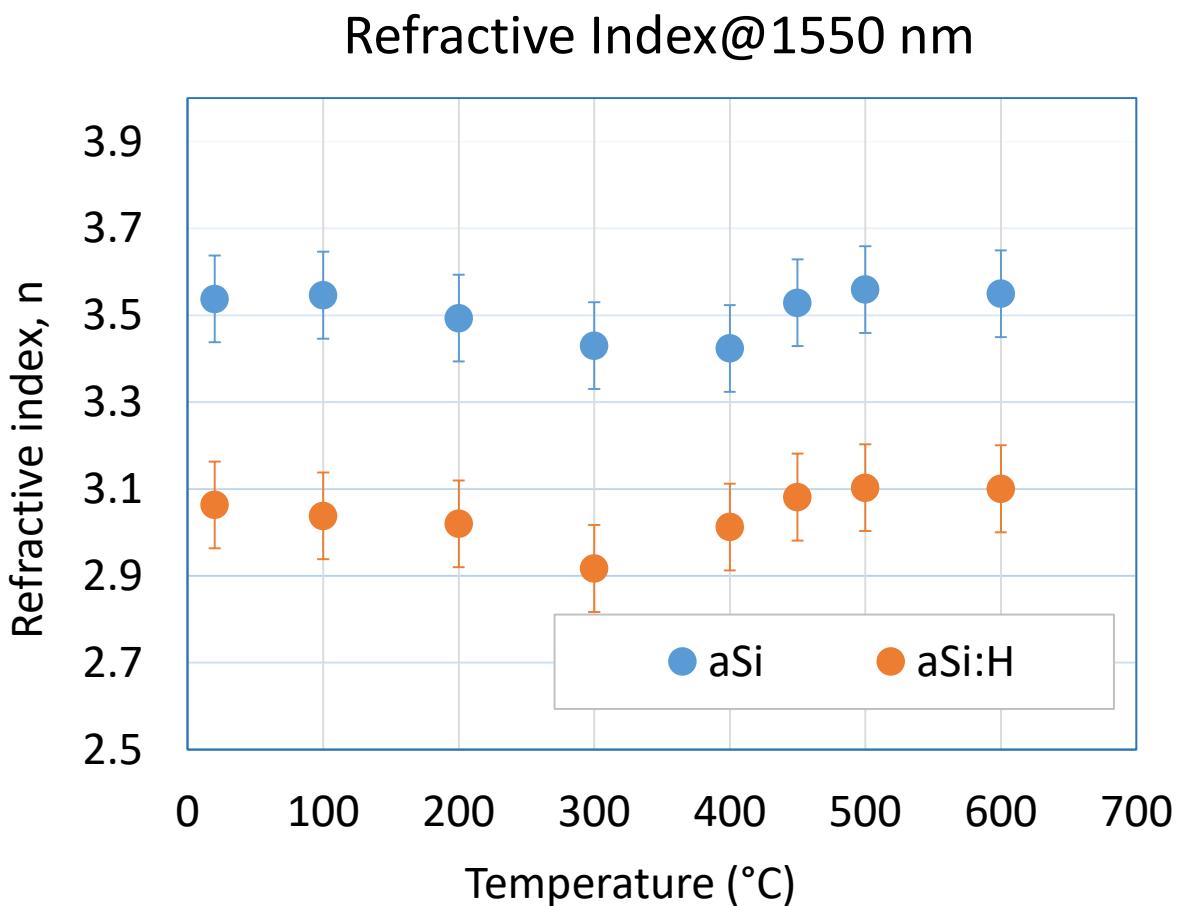
measured with the help of a Cary 5000 spectrophotometer.

## Optical properties

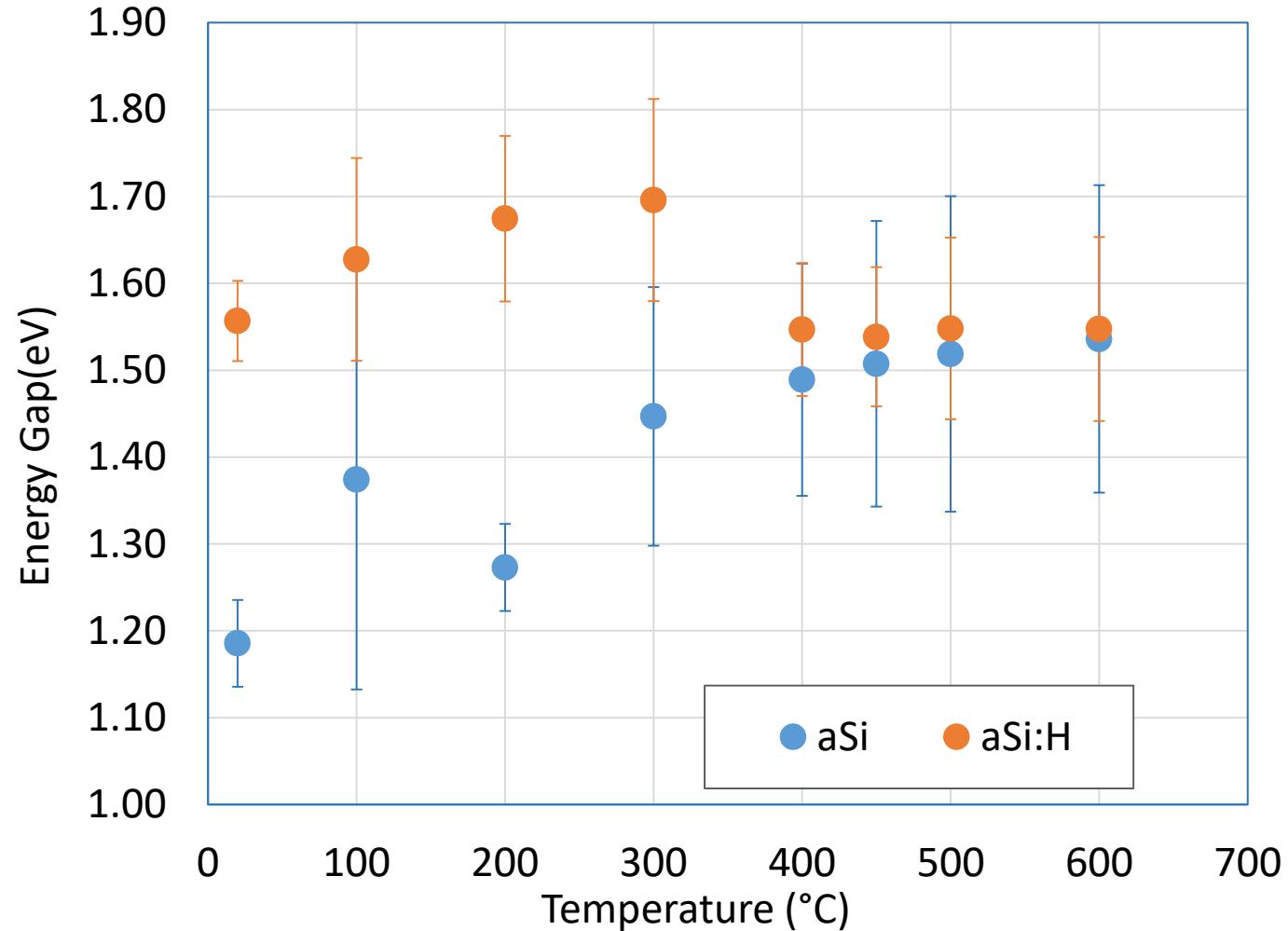
analyzed using two Kramers-Kronig consistent models:  
Tauc-Lorentz and OJL models.  
*(both are considered for final results)*



# Refractive Index and thickness vs heat treatment



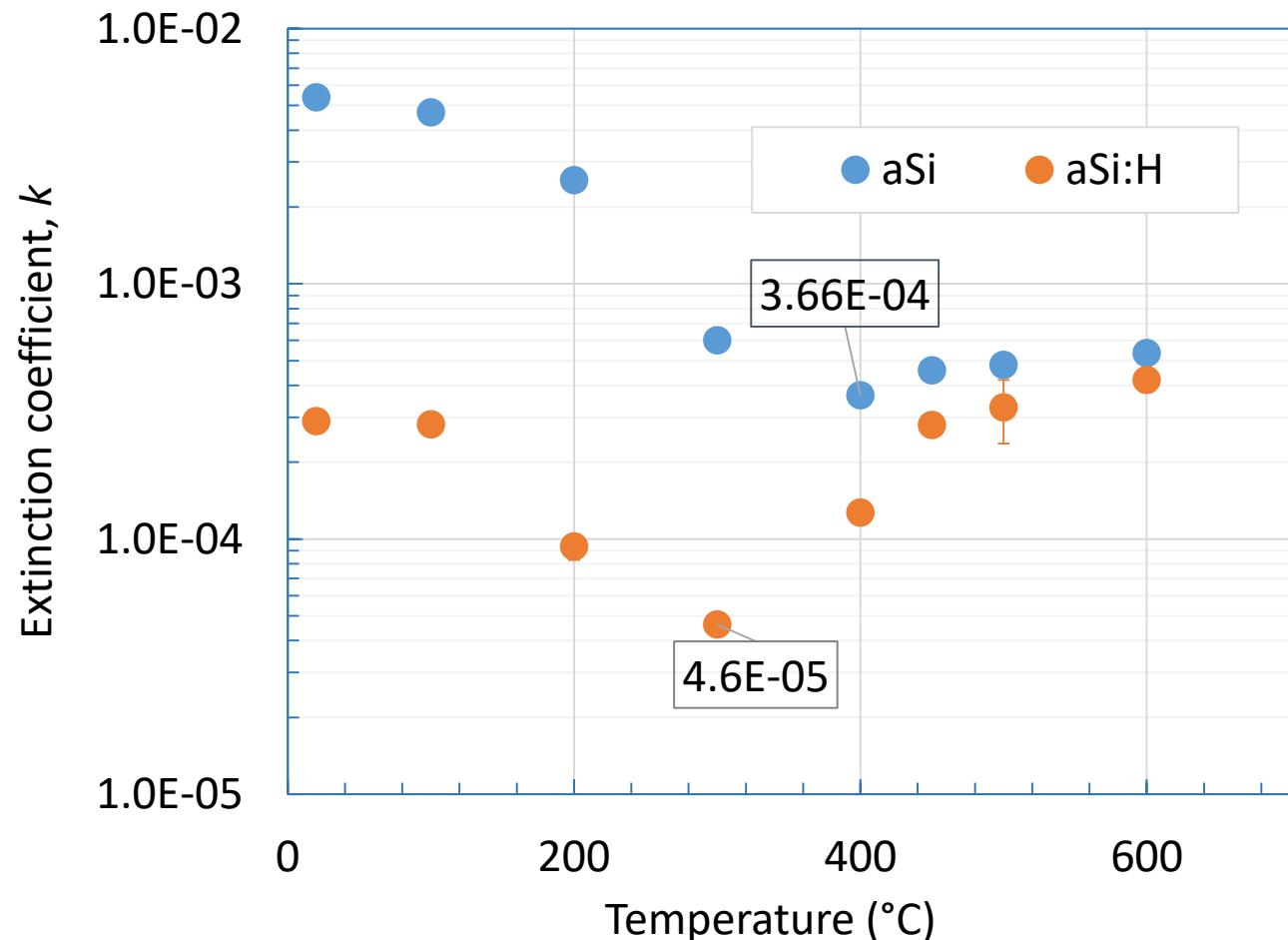
# Energy gap vs heat treatment



# Extinction Coefficient @ 1550 nm

## Results obtained in our study

1. Absorption measurements done with PCI (photo thermal common path interferometer) technique.
2. Absorption values were used to extract extinction coefficient values with the help of Tfcalc software.



## Results from other studies conducted before in LVK

<https://dcc.ligo.org/LIGO-G2400542> (LVK March 2024)

Phys. Rev. Lett. 131, 256902

# RBS and ERDA analysis

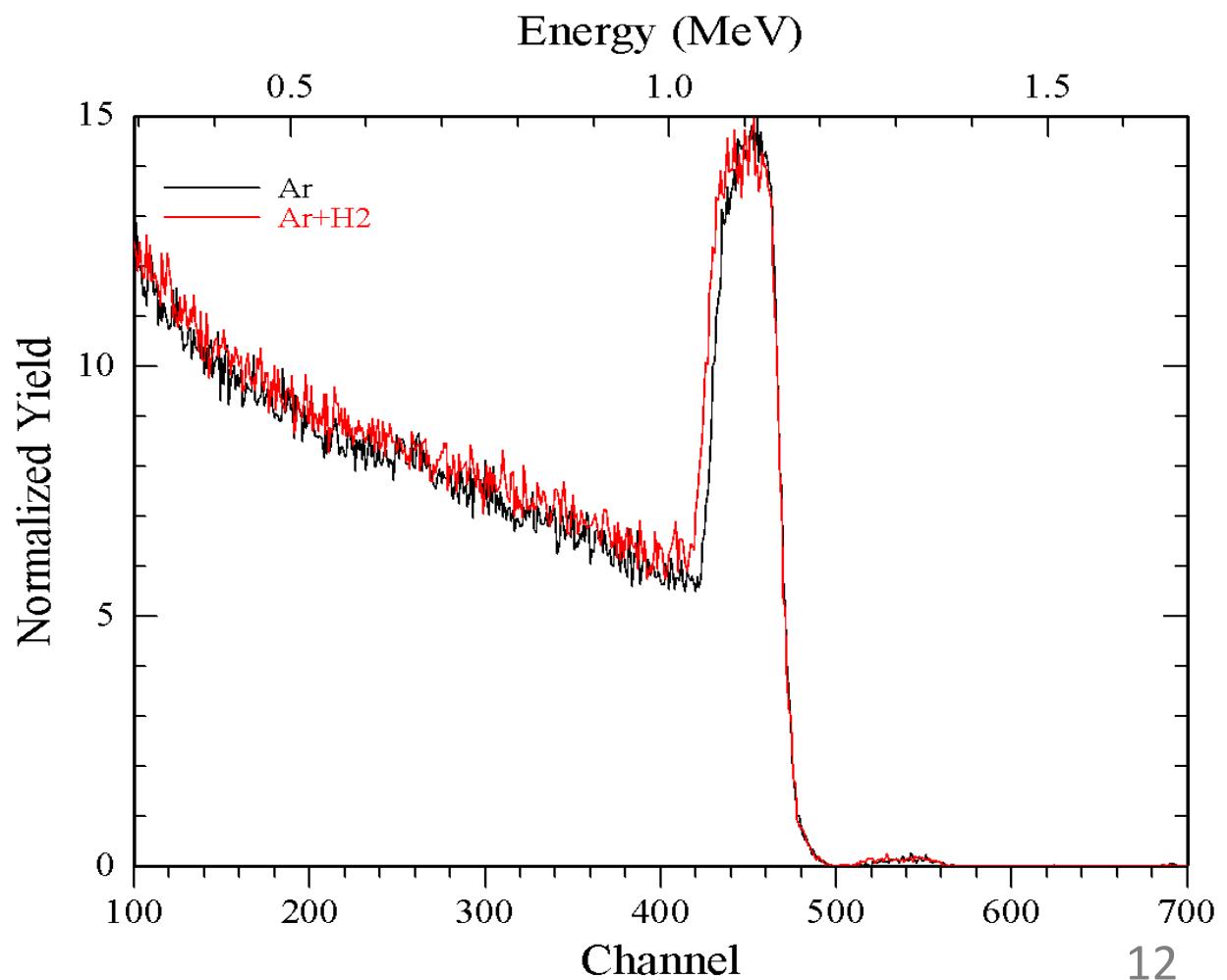
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RBS - Rutherford backscattering spectrometry

ERDA - Elastic recoil detection analysis

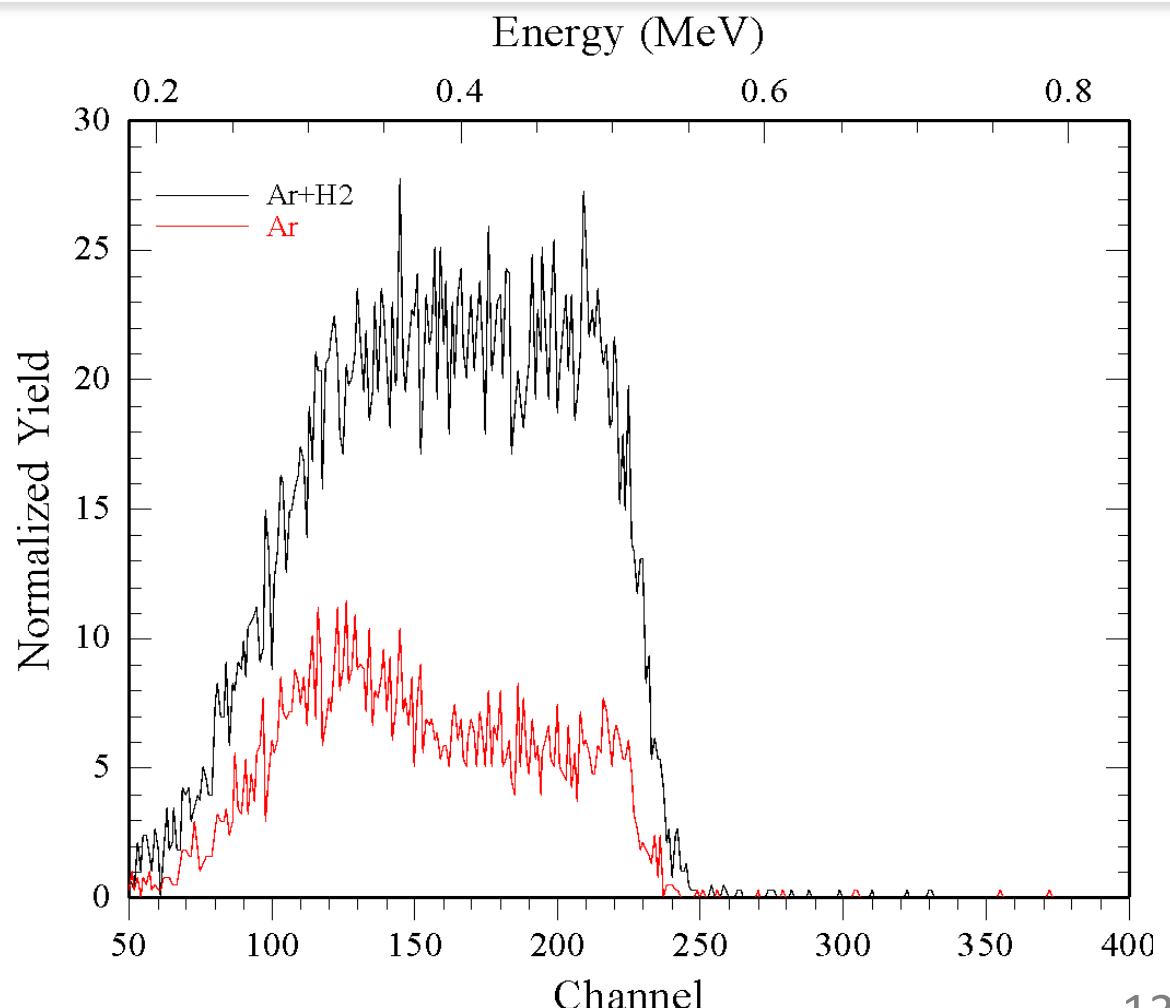
# RBS measurements

Film Atomic Combination		
Sample	Si dose (atom cm <sup>-2</sup> )	Film Thickness
aSi	9.3e17	191
aSi:H	9.7e17	215

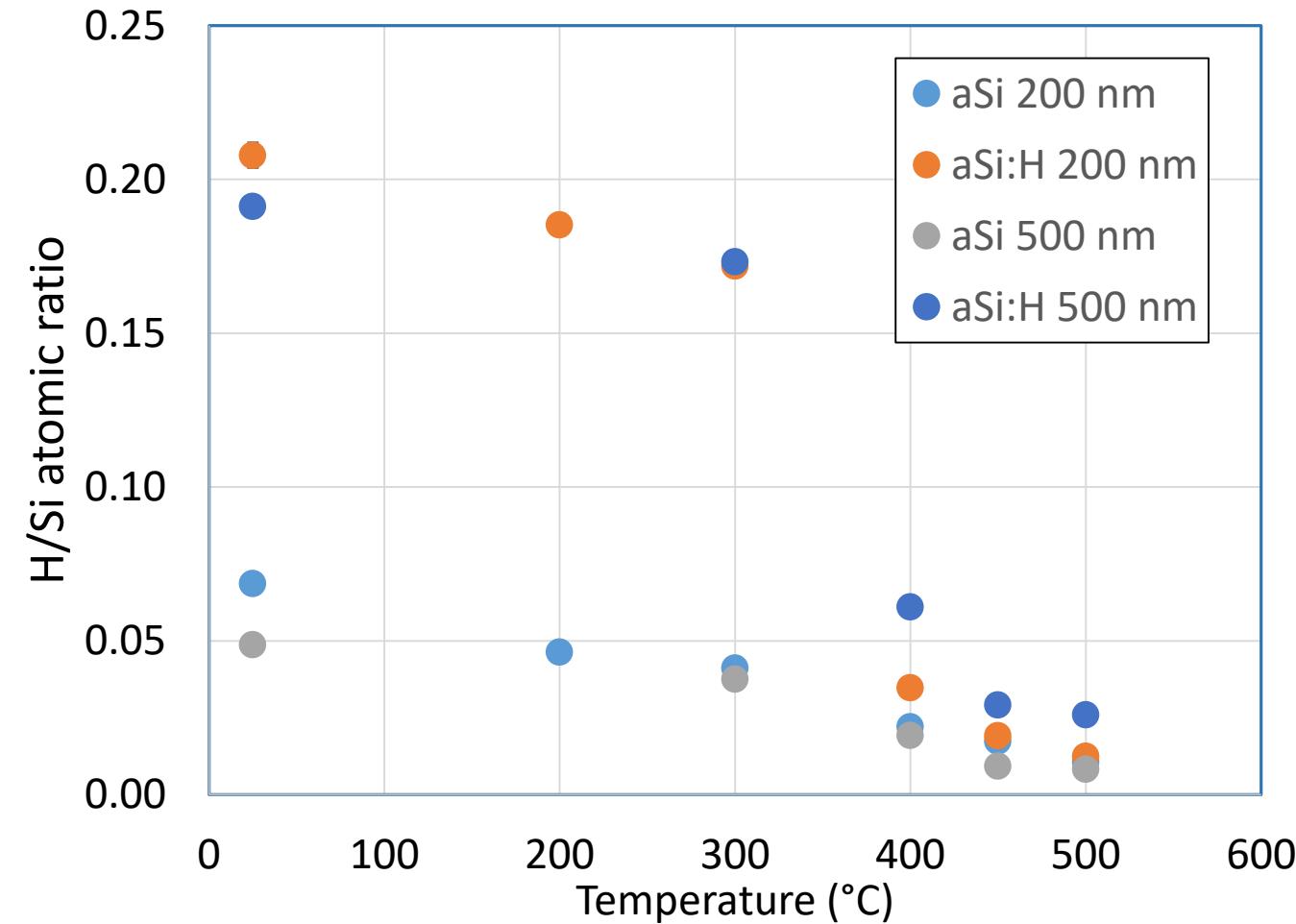


# ERDA Measurements

Film Atomic Combination		
Sample	H/Si	Ar/Si
aSi	0.07	0.004
aSi:H	0.21	0.006



# RBS+ERDA after annealing

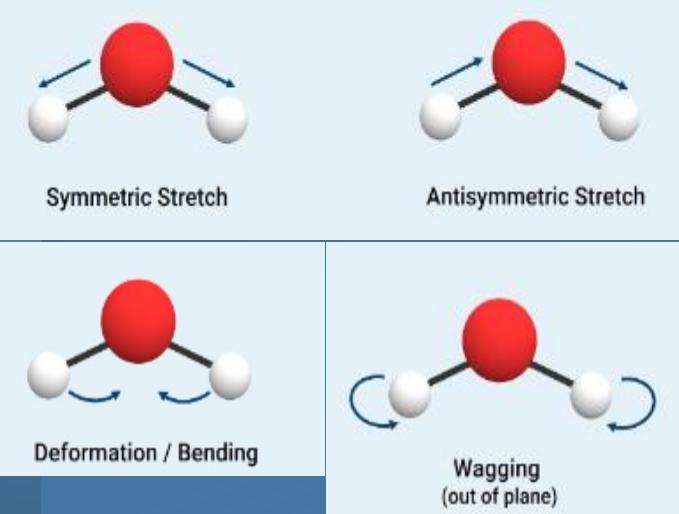
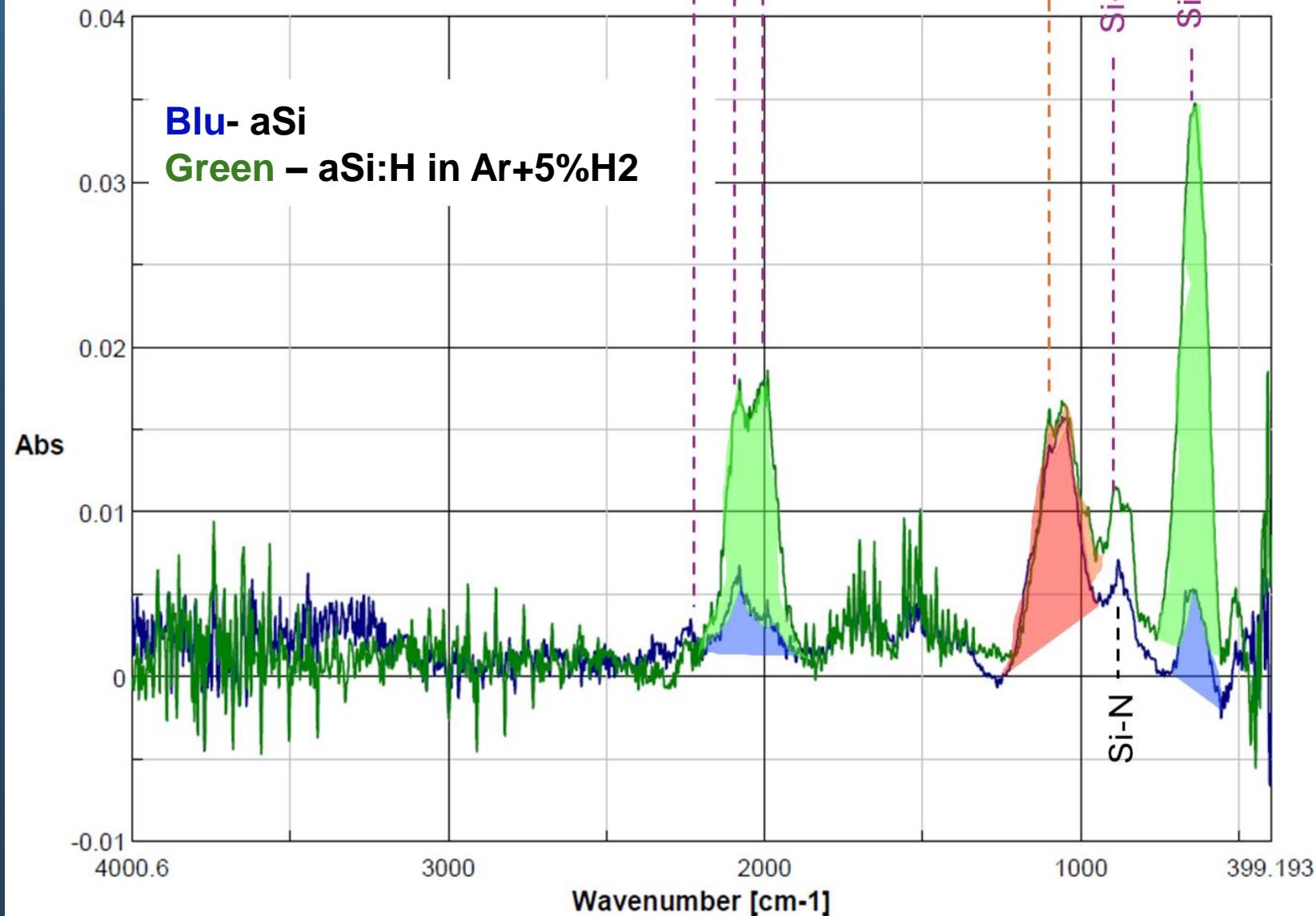


H/Si atomic ratio obtained by ERDA+RBS has similar **trend** in respect to Si-H bonds concentration obtained by FTIR

There is **significant decrease** of H/Si atomic ratio in aSi 937 sample sputtered in 5%H<sub>2</sub>+Ar after 300°C

# FT/IR analysis

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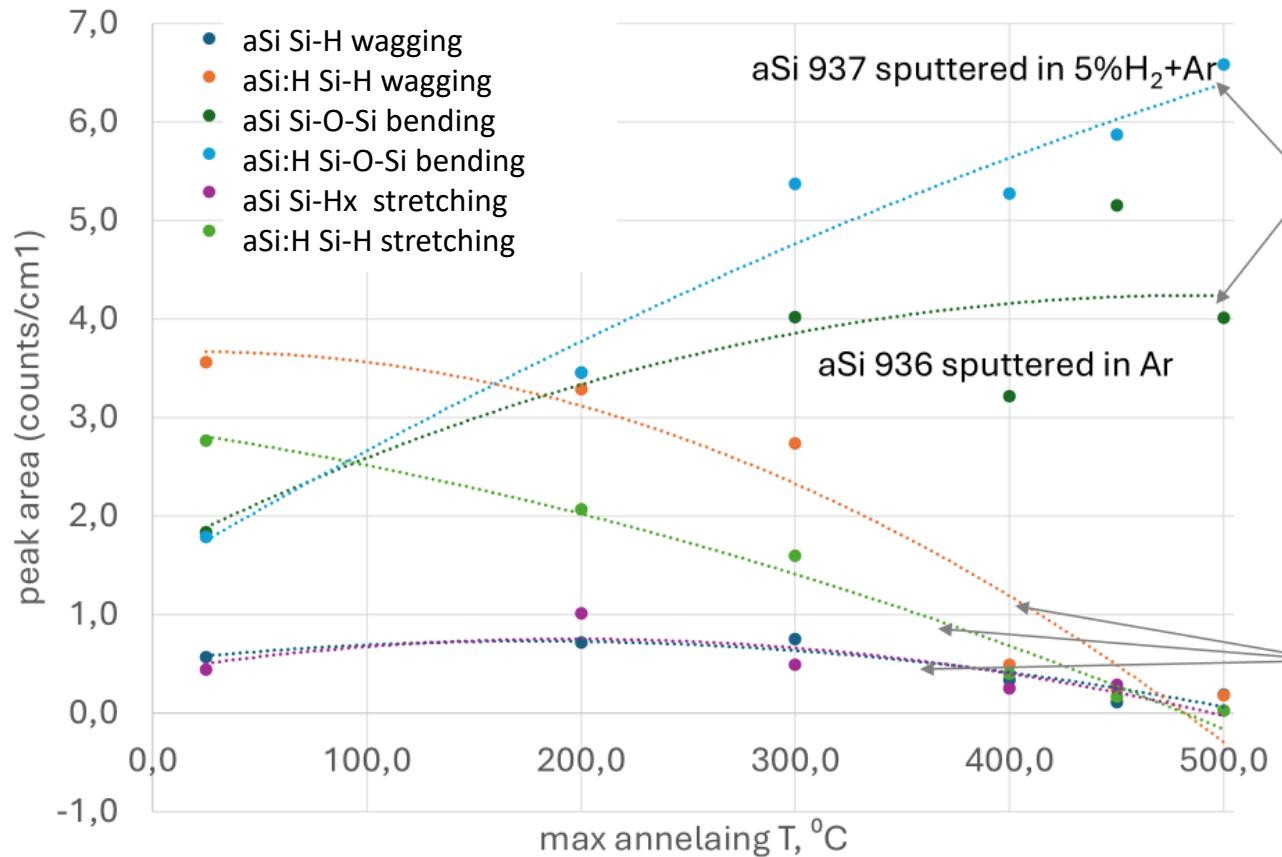


Measuring area under the peaks correlated to

- Si-H<sub>2</sub> Si-H stretching modes
- Si-H<sub>x</sub> wagging mode
- Si-O-Si single or multiple band

Proportional to “concentration” of the bond in the film

# FTIR: peak area evolution (trend) on annealing

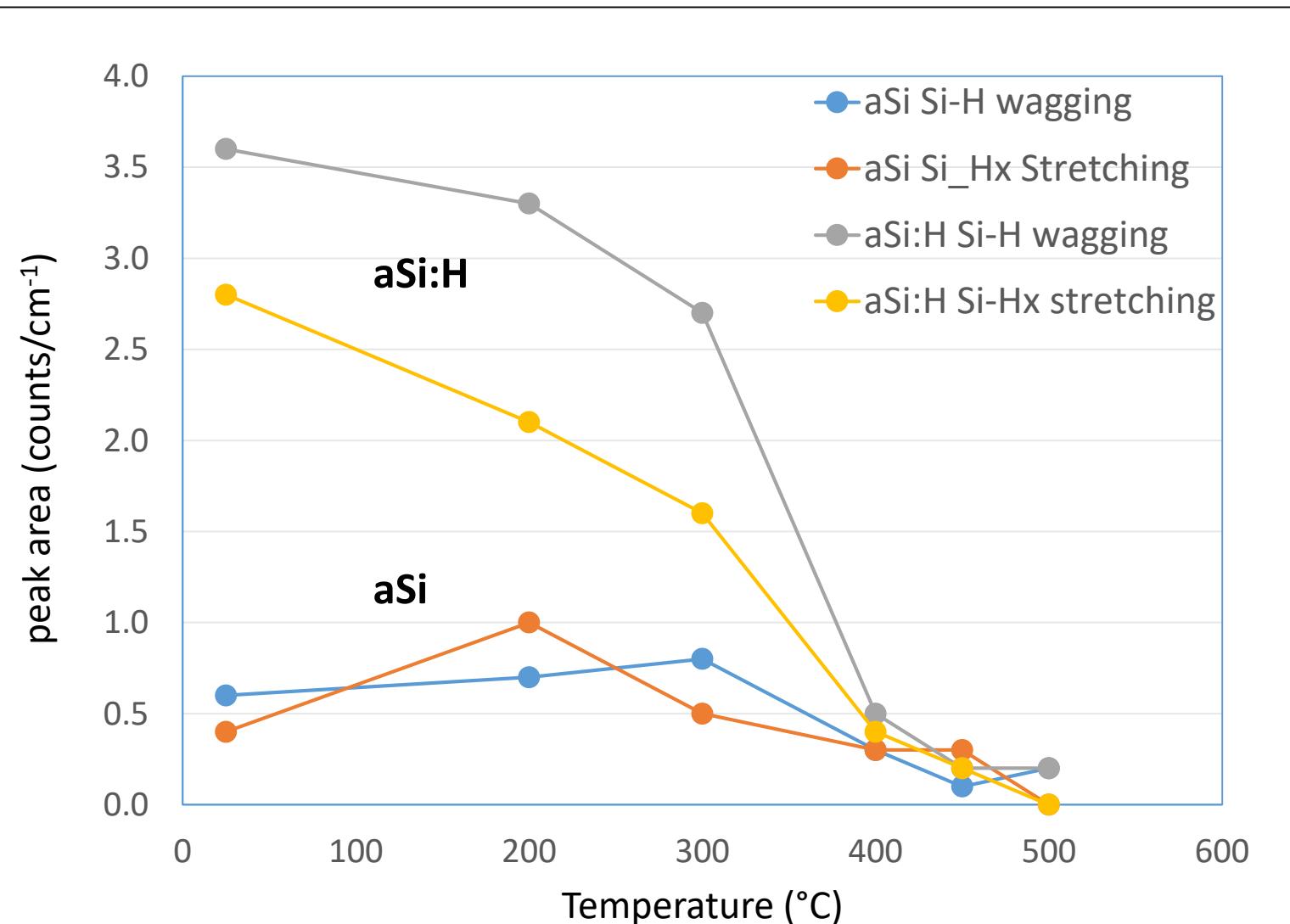


Si-O-Si bonds “concentration” increase  
=  
oxidation

We observe that aSi 937 deposited in 5%H<sub>2</sub>+Ar is more oxidized on annealing than aSi 936 deposited in Ar

Si-H bonds “concentration” decrease  
=  
Si-H bonds decomposition,  
=  
H content decrease

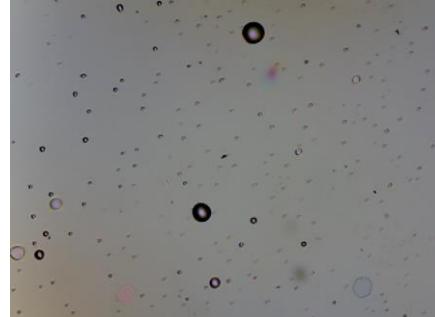
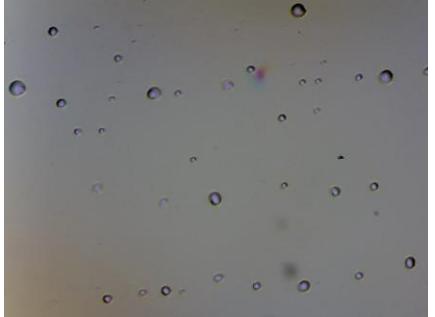
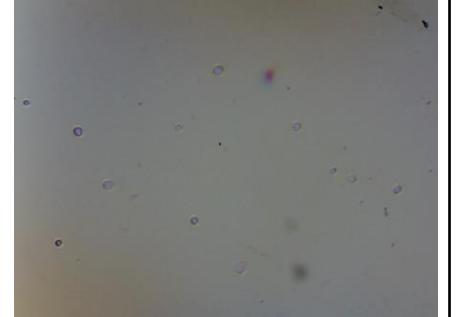
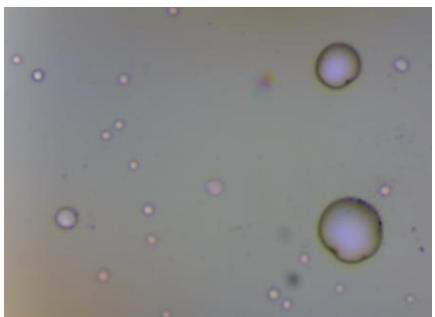
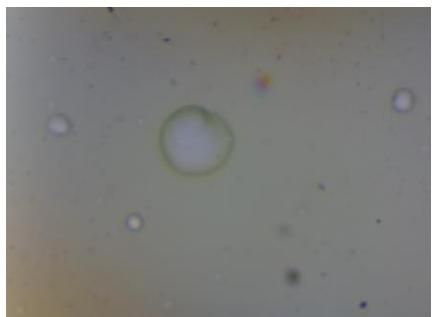
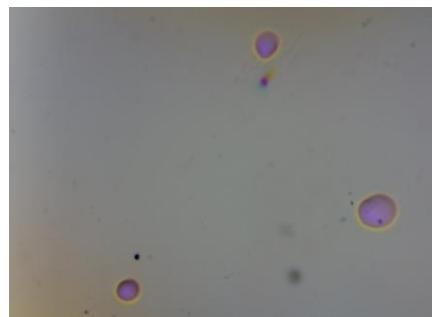
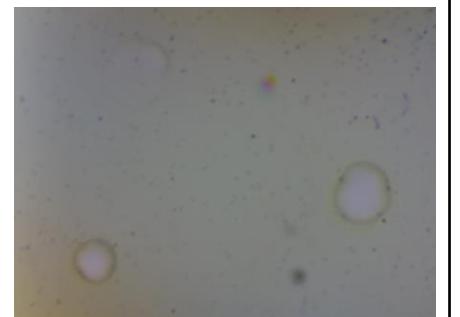
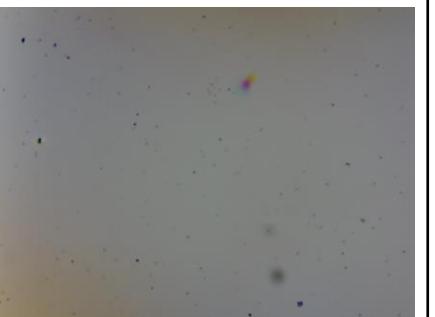
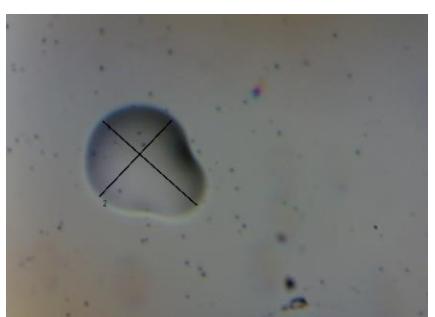
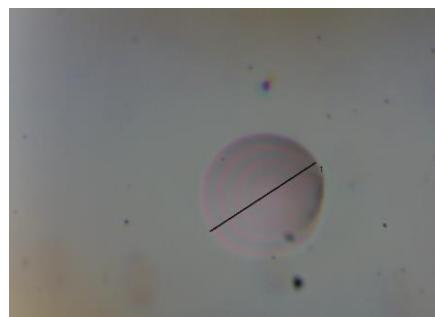
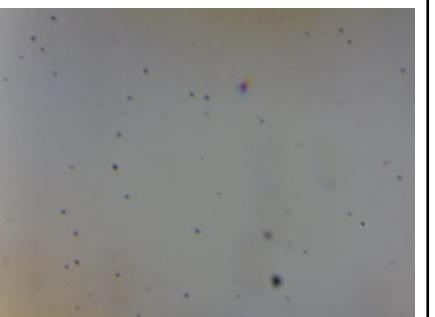
# FTIR: peak area evolution (trend) on annealing



It is evident the change of the trend of hydrogen release (decomposition of Si-H bonds) in aSi 937 sample sputtered in 5%H<sub>2</sub>+Ar

In aSi 936 sample sputtered in pure Ar the release of hydrogen on increasing annealing temperature seems to have a softer trend

**After annealing at 450°C all samples show blistering. Ar deposited samples have more bubbles**

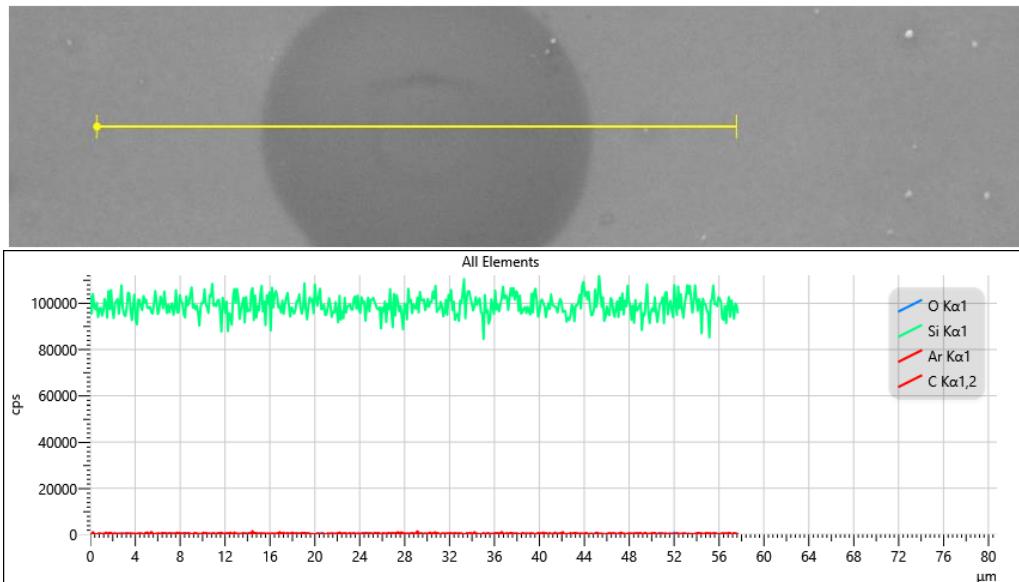
Opt. mic. magnifica tion	Ar		Ar+5% H <sub>2</sub>		
	200 nm	500 nm	200 nm	500 nm ( set 1)	500 nm ( set 2)
x5					
x20					
x50					

# SEM\_EDS measurement of bubbles

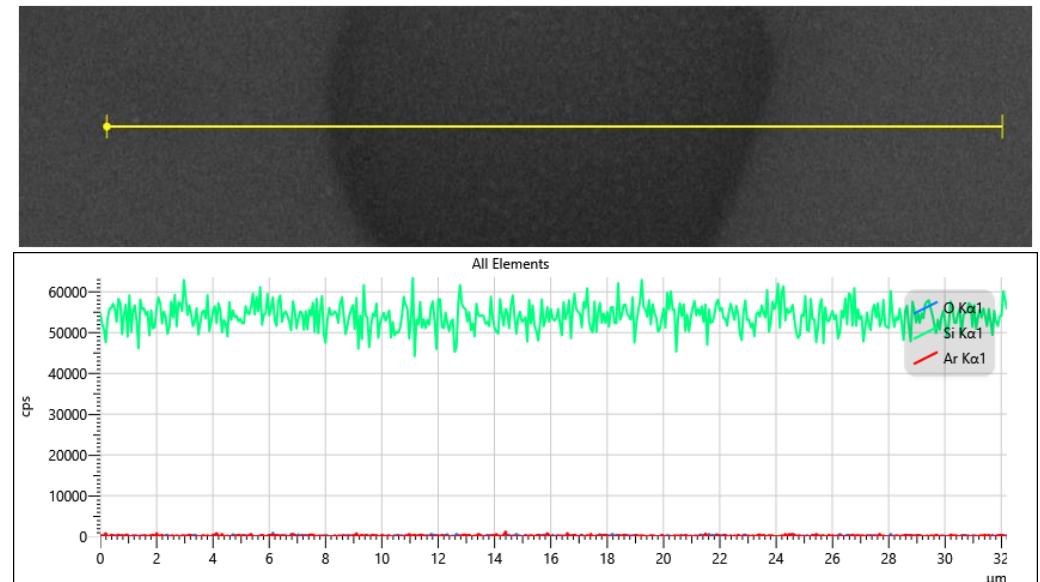
*Hypothesis: bubble is full of Ar*

**30kV e-beam**

aSi\_ 300°C, 400°C, 450°C, 500°C



aSi:H\_ 300°C, 400°C, 450°C, 500°C



No Ar variation through line scan was observed

# Summary

The lowest extinction coefficient was observed for magnetron sputtered aSi:H (5% H<sub>2</sub>) at 300 °C was  
4.6E-05



It is almost a factor of 12 times with the non hydrogenated sample at the same temperature and 8 times to the lowest at 400 °C



The hydrogen content rapidly falls after 300 °C observed from the ERDA measurements . It is also observed that the Si-H bond “concentration” from area under the peaks in FT/IR analysis also seems to decrease.



Lower refractive indices observed for hydrogenated samples ( 3.01 (aSi:H) and 3.42 (aSi) at 300 °C)  
(12% less than aSi at 300 °C)

# Further steps for amorphous silicon

Measure 500 nm thick single layers for characterization



Development of multiple layer stack of amorphous silicon and silicon nitride and it's study

Thank you for your  
attention!

